

# Thin films growth and applications

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**Outline** 

- Why thin films?. Motivation
- Growth techniques
- Deposition mechanisms
- Epitaxy
- In-situ characterization: RHEED
- Examples

## Why thin films?

 Possibility of growth metastable phases, difficult or impossible to obtain in bulk/single crystals

 Possibility of obtain materials with improved properties (strain effects, finite size effects)

Combine different materials in multilayers/composites, with improved functionalities

This is the form needed for applications



## Thin Film Applications

Thin-film property category	Typical applications						
Optical	Reflective/antireflective coatings						
	Interference filters						
	Decoration (color, luster)						
	Memory discs (CDs)						
	Waveguides						
Electrical	Insulation						
	Conduction						
	Semiconductor devices						
	Piezoelectric drivers						
Magnetic	Memory discs						
Chemical	Barriers to diffusion or alloying						
	Protection against oxidation or corrosion						
	Gas/liquid sensors						
Mechanical	Tribological (wear-resistant) coatings						
	Hardness						
	Adhesion						
	Micromechanics						
Thermal	Barrier layers						
	Heat sinks						

#### TABLE 1.1 Thin-Film Applications

## Thin film process steps



## Vacuum technology



Figure 3.1 Typical vacuum-system components for thin-film deposition.

## Vacuum technology

Pressure ranges	Name	Category	Approx. \$/(l/s)	Backing pump req'd?	Oil present?		Ducklamatia	011	
					Inlet	Outlet	gases and vapors	comments	
<u>}·+−−+</u> +	Dry rotary	Displacement	1000	No	No	Yes	Condensables require gas ballasting; see text		
ultimate limits	Oil-sealed rotary		300	No	Yes	Yes		Common for roughing/backing	
	Roots blower		70	Yes	No	Yes		Oil contam. unless foreline purged	
I.I.	Molecular drag		35	Yes	No	Yes*	Low compression ratio for H <sub>2</sub> and He		
←───	Turbo-molecular		40	Yes	No	Yes*			
◄	Oil diffusion		5	Yes	Yes	Yes		Greatest risk of oil contam.	
}{	Cryosorption	Trapping	450	No	No	(No outlet)	Explosion danger with flammables	For dry roughing	
◄	He-cycle cryopump		7	No	No	No <sup>†</sup>		Low capacity for He, H <sub>2</sub>	
◄{	Sputter-ion		25	No	No	(No outlet)	Poor for inerts		
molecular <del>dia in</del> fluid transition	Flow regime for 5 cm diameter tube								
IIIIIIIIIIII I -5 I 0 I +5 Pa -10 -5 0 Torr 4 I atm	log of pump inlet pressure								

## Thin film techniques

**Physical Vapor Deposition (PVD)** 

- Thermal evaporation
- Electron beam evaporation
- Sputtering
- Molecular beam epitaxy
- Pulsed laser deposition

**Chemical Vapor Deposition (CVD)** 

Other (spin coating, dip coating, atomic layer deposition...)

#### Thin film techniques: Thermal evaporation



- The coating material is evaporated in vacuum with the help of resistive heating.
- Advantages: Good purity and good surface quality
- Disadvantages: non-uniformity of the deposit, poor adherence sometimes, evaporation of complex compounds is problematic (different evaporation rates)

#### Thin film techniques: Electron beam evaporation



- A target anode is bombarded with an electron beam given off by a charged tungsten filament under high vacuum. The electron beam causes atoms from the target to transform into the gaseous phase. These atoms then precipitate into solid form, coating everything in the vacuum chamber with a thin layer of the anode material
- Multiple electron guns can be used simultaneously in a single system.

## Thin film techniques: Sputtering



- Atoms are ejected from a solid target material due to bombardment by energetic particles.
- The primary particles for the sputtering process can be supplied in a number of ways, for example by a plasma or an ion source.
- Sputtering sources often employ magnetrons that utilize strong electric and magnetic fields to confine charged plasma particles close to the surface of the sputter target.
- Charge build-up on insulating targets can be avoided with the use of RF sputtering.
- Advantages: good uniformity in thickness, good adhesion, stochiometry preservation

#### Thin film techniques: Molecular beam epitaxy





- Intricate structures of layers of different materials may be fabricated this way.
- Typically used to grow high quality semiconductors (i.e. solar cells)
- More recently has been used to deposit oxide materials for advanced electronic, magnetic and optical applications. For these purposes, MBE systems have to be modified to incorporate oxygen sources.

#### Thin film techniques: Molecular beam epitaxy





• MBE takes place in ultra-high vacuum (10<sup>-8</sup> Pa). The deposition rate (typically less than 3000 nm per hour) allows the films to grow epitaxially. The absence of carrier gases as well as the ultra high vacuum environment result in the highest achievable purity of the grown films.

• In solid-source MBE, elements in ultra-pure form, are heated in separate Knudsen effusion cells until they sublime. The gaseous elements then condense on the wafer, where they may react with each other. The evaporated atoms do not interact with each other or vacuum chamber gases until they reach the wafer, due to the long mean free paths of the atoms.

## Thin film techniques: Chemical Vapor Deposition

![](_page_13_Figure_1.jpeg)

• Reactive gaseous precursos are brought onto the substrate at high temperatures. The reagents are decomposed and adsorbed, and the film is formed as a result of the chemical reaction between the precursos. By products are desorbed from the substrate and removed by a gas flow.

- Advantages: good quality films, uniformity in thickness.
- Disavantages: high temperature of the substrate

#### History:

- First experiments in the 1960's
- Limited efforts in the 1970's and 1980's
- Boom in late 1980's → thin films of HTS
- Today, PLD is used for:
  - insulators
  - semiconductors
  - metals
  - polymers
  - biological

![](_page_14_Figure_11.jpeg)

#### Advantages:

- Congruent deposition  $\rightarrow$  highly non-equilibrium process
- Deposition under reactive gases
- Multilayer growth

#### **Deposition parameters:**

- Laser fluence  $\rightarrow$  minimum threshold to avoide heating/evaporation
- Laser wavelenght  $\rightarrow$  UV, good absorption
- Gas pressure during growth  $\rightarrow$  participation on the chemistry of film growth + control of the kinetic energy of the ablated species
- Substrate temperature  $\rightarrow$  difussion of species on substrate surface
- Target-substrate distance  $\rightarrow$  growth rate, particulate

<u>Targets</u>  $\rightarrow$  high optical absorption at laser wavelenght

In general, only <u>cation</u> stochiometry matching is required

Ceramics (dense!!), single crystals, soft materials (bio)  $\rightarrow$  material of interest embedded in a matrix of optically absorving material

Reactive PLD  $\rightarrow$  metal targets  $\rightarrow$  low efficiency process (metals reflect!) + particulate due to molten droplets. Solution: liquid metal targets.

Growth rate: 1 pulse  $\rightarrow$  usually 1 sub-monolayer  $\rightarrow$  layer-by-layer growth monitored by RHEED

**Deposition rates: 0.001 to 1Å/pulse** 

Scaling  $\rightarrow$  industry requires large areas  $\rightarrow$  manipulation of plume-substrate positioning  $\rightarrow$  substrate rotation + rastering of ablation beam over a large target area

<u>Drawbacks</u>  $\rightarrow$  changes in plume energy and stochiometry as one moves to the edge of the plume

<u>Volatile species</u>  $\rightarrow$  non-stochiometric films due to evaporation  $\rightarrow$  cation excess in the target or mosaic target

![](_page_17_Figure_4.jpeg)

#### Main disadvantage:

• Particulate formation (problem for multilayers!) → large penetration depth of the laser inside the target. High density targets should be used!

![](_page_18_Picture_3.jpeg)

LCMO/Si 6 Ton

LCMO/Si 10 Ton

• Mechanical techniques can minimize particulate formation  $\rightarrow$  velocity filters, off-axis laser deposition

 Nucleation during flight can also produce particulate → background pressure sufficiently high for particle nucleation. Substrate-target distance is also important.

![](_page_19_Figure_3.jpeg)

Figure 1.4 Cross-section TEM image of a CeO<sub>2</sub> film grown at high and low pressure, with CeO<sub>2</sub> nanoparticles forming at the high background pressure.

• Different types of particulate

![](_page_20_Picture_2.jpeg)

LCMO/Si

**Applications of PLD** 

• Complex oxides thin films growth  $\rightarrow$  stochiometry transfer is critical!  $\rightarrow$  YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7</sub>, manganites A<sub>x</sub>A'<sub>1-x</sub>MnO<sub>3</sub>

Even more sophisticated oxides  $\rightarrow$  magnetoplumbite Ba<sub>2</sub>Co<sub>2</sub>Fe<sub>12</sub>O<sub>22</sub>  $\rightarrow$  extremely complex crystal structure

![](_page_21_Figure_4.jpeg)

Figure 1.6 Crystal structure, X-ray diffraction data, and TEM image of a Ba<sub>2</sub>Co<sub>2</sub>Fe<sub>12</sub>O<sub>22</sub> film grown by PLD [Ohkubo et al., 2003].

**Applications of PLD** 

• Epitaxial interfaces and superlattices, even chemically dissimilar materials (i.e. metal-oxide  $CeO_2/Ge$ )  $\rightarrow$  investigation of reduced dimensionality effects

 $KTaO_3/KNbO_3 \rightarrow individual layers as thin as 1 u.c. \rightarrow sharp interfaces (atomic scale). PLD is competitive with MOCVD and MBE$ 

![](_page_22_Figure_4.jpeg)

Figure 1.7 Cross-section Z-contrast STEM image of a KTaO<sub>3</sub>/KNbO<sub>3</sub> superlattice structure grown by pulsed laser deposition.

**Applications of PLD** 

 Superconducting Electronic Devices → high freq. electronics for RF/microwave communications and SQUIDS

• Supercoduncting wires for viable HTS technology. RABiTS  $\rightarrow$  substrate tape with appropriate texture + epitaxial YBCO growth. The integrity of the epitaxial layer should be maintended allong the conductor (~kms!!).

![](_page_23_Figure_4.jpeg)

Figure 1.11 Schematic of the RABITS process for HTS-coated conductor fabrication.

**Applications of PLD** 

Novel oxide devices concepts

• High-k dielectrics for Si microelectronics (miniaturization  $\rightarrow$  replacement of SiO<sub>2</sub>)  $\rightarrow$  increase of gate capacitance in transistors  $\rightarrow$  epitaxial, polycristalline and amorphous dielectrics (i.e. HfO<sub>2</sub>)

• Memory devices based on FE  $\rightarrow$  Au/PZT/Bi<sub>4</sub>Ti<sub>13</sub>O<sub>12</sub>/p-Si diode  $\rightarrow$ memory is accomplished via polarization hysteresis related to FE switching

![](_page_24_Figure_5.jpeg)

Figure 1.13 Capacitance versus bias voltage characteristics of Au/PZT/BIT/p-Si diode between -4.5 and 4.5 V bias voltage at a bias frequency of 100 kHz. The thickness of PZT and BIT were 300 and 100 nm, respectively [Yu et al., 2001].

#### **Oxide sensor devices**

• Chemical detectors based on semiconducting oxides (i.e. gas sensors, electronic noses  $\rightarrow$  SnO<sub>2</sub> for NO<sub>2</sub> detection). Uncooled bolometers for infrared detection (VO<sub>2</sub>, I-M transition and high temperature coefficient).

![](_page_25_Figure_3.jpeg)

Figure 1.21 Microbolometer element, without air-gap thermal isolation, was fabricated using pulsed laser deposited vanadium oxide as the IR sensing layer [Kumar et al., 2003].

![](_page_25_Figure_5.jpeg)

Resistance (ohms)

Protective coatings and barriers  $\rightarrow$  borides, nitrides, carbides. Diamond-like carbon can deposited by PLD.

• Biocompatible coatings  $\rightarrow$  dental and orthopedic implants (hydroxilapatite, calcium orthoposphate), similar to bone tissue

Particle coatings for drug delivery → Budenoside (asthma treatment) covered with a polymer

![](_page_26_Figure_4.jpeg)

Nanosystems synthesis: nanocrystals, nanowires, nanocomposites for nanotransistors, nanosensors study of quantized electronic systems.

For example, an hybrid PLD/CVD was used to synthesize NW with periodic composition

![](_page_27_Figure_3.jpeg)

![](_page_27_Figure_4.jpeg)

Figure 1.24 Film growth process that is a hybrid of pulsed laser ablation and chemical vapor deposition was used to synthesize semiconductor nanowires with periodic longitudinal heterostructures [Wu et al., 2002].

Figure 1.25 Cross-sectional transmission electron microscopy image of a single-crystal compositionally modulated nanowire with a Si/SiGe superlattice structure along the length of the wire [Wu et al., 2002].

Polymer and organic thin films

Teflon can be deposited on metallic surfaces by PLD (protective coatings for bio applications: metal implants, tooth fillings, jewerly). PMMA can also be deposited by PLD. Effective PLD should avoid organic molecules decomposition. Also, light-induced descomposition should be avoid during ablation  $\rightarrow$  a photosensitizer is added to the target  $\rightarrow$  excitation of the latter

DNA thin films can be grown by PLD, for microfluidic biosensors, biocompatible coatings, gene recognition microarrays.

![](_page_29_Figure_1.jpeg)

## **Typical PLD chamber**

#### FI-UBA: home made PLD system

![](_page_30_Picture_2.jpeg)

#### **CAC-CNEA: NBM PLD commercial system**

![](_page_30_Picture_4.jpeg)

![](_page_31_Picture_0.jpeg)

- (1) Arriving atoms are adsorbed on the substrate surface
- (2) Diffuse some distance before becoming incorporated into the film
- (3) Reaction to form the bonds (with other species + surface)
- (4) Nucleation (initial agregation of the film)
- (5) Development of a structure or morphology, including topography and crystallography
- (6) Diffusional interactions within the bulk of the film and with the substrate

![](_page_32_Picture_0.jpeg)

- Physisorption (van der Waals interaction)
- Chemisorption (chemical bonds are created)

![](_page_32_Figure_3.jpeg)

 $\gamma \rightarrow surface energy$  per unit area of surface (N/m)

Surface diffusion  $\rightarrow$  minimization of surface energy

 $\gamma$  depends on chemical composition, crystallographic orientation, atomic-scale roughness, etc.

 $\gamma$  is usually <u>anisotropic</u>  $\rightarrow$  low energy facets

For deposition onto a foreign substrate, one must take into account  $\gamma_S$  (substrate free surface),  $\gamma_I$  (substrate-film interface) and  $\gamma_F$  (film free surface).  $\gamma_I$  is tunable  $\rightarrow$  decreases when film-substrate bonding increases

# (a)layer-by-layer growth (Frank-van der Merwe) if γ<sub>F</sub>+γ<sub>I</sub><γ<sub>S</sub> (surf. energy is lower for the wetted substrate than for the bare one) (b) 3D-islands (Volmer-Weber)

(c) Stranski-Krastanov (Elastic energy is accumulated as the film grows 2D. At some critical height, the free energy of the film can be lowered if the film breaks into isolated islands, where the tension can be relaxed laterally.)

![](_page_34_Figure_3.jpeg)

Epitaxy occurs when the bonds of the film crystal align with the bonds of the substrate surface, making the interfacial energy  $\gamma_{\rm I}$  very small. So, it is energetically favourable for the film material to crystallography align itself with the substrate

• Structural matching with the substrate

$$\frac{\Delta a}{a} = \frac{(a_{pel} - a_{subs})}{a_{subs}} * 100$$

• Substrate temperature window (interdifusion with the substrate, difussion on the surface)

Epitaxy improves interface quality in multilayered structures and results in films with few grain boundaries

![](_page_36_Picture_0.jpeg)

Epitaxy is particulary sensitive to degradation by impurities and defects. Even one monoloayer of disordered contaminant disrupts epitaxy.

![](_page_36_Figure_2.jpeg)

Substrate requirements to obtain epitaxy: crystallographic order, sub-monolayer surface cleanless and chemical inertness toward the depositing species.

## **Kinetics of thin film growth**

![](_page_37_Picture_1.jpeg)

Kinetic effects may play a role in adatom incorporation:

- (a) <u>Downhill funneling</u>: the deposited particle slides down a slope until a local minimum of the surface height is reached
- (b) <u>Knockout process</u>: the momentum of the arriving particle suffices to push out a surface adatom at a terrace edge
- (c) <u>Steering effect</u>: attractive forces can influence the trajectory of the arriving particle significantly

**RHEED:** Reflection High Energy Electron Diffraction

In-situ characterization → Electron diffraction in grazing incidence: surface effect (a few nm)

**Electrons energy ~ 10-50keV** 

![](_page_38_Figure_4.jpeg)

Figure 4.1 Schematic view of the RHEED geometry.  $\Theta_I(\Theta_F)$  and  $\phi_I(\phi_F)$  are the incident and azimuthal angles of the incident (diffracted) beam.  $R_S$  is the distance between substrate and phosphor screen and S the distance between the diffraction spots or streaks.

Electrons energy ~ 10-50keV  $\lambda(\text{\AA})$ =(150/E)<sup>1/2</sup> ~ 0.05-0.1 Å

Incidence angle ~ 0.1°-5°

![](_page_38_Picture_8.jpeg)

**RHEED:** Reflection High Energy Electron Diffraction

Weakly interacting diffraction techniques  $\rightarrow$  kinematic scattering theory RHEED  $\rightarrow$  strong interactions!  $\rightarrow$  no quantitative description

![](_page_39_Figure_3.jpeg)

Figure 4.2 Ewald sphere construction in (a) three dimensions and (b) a section of the horizontal z = 0 plane.

Reciprocal lattice of a 2D surface  $\rightarrow$  lattice of thin rods, perpendicular to the surface. High electronic energy  $\rightarrow$  Ewald sphere is large  $\rightarrow$  few sport are seen

 $K_{s}-K_{0} = G$ 

**RHEED:** Reflection High Energy Electron Diffraction

#### Kikuchi lines → multiple scattering

![](_page_40_Picture_3.jpeg)

Figure 4.3 Typical RHEED pattern as recorded from a perfect SrTiO<sub>3</sub> surface.

#### Clear and sharp Kikuchi lines $\rightarrow$ indication of flat and crystalline surface

**RHEED:** Reflection High Energy Electron Diffraction

**Utility of RHEED** 

• Determination of lattice parameters

 $n/d_x = 1/\lambda(\cos\theta_f - \cos\theta_i)$  $n/d_v = 1/\lambda(\cos\theta_f * \sin\phi_f)$ 

![](_page_41_Figure_5.jpeg)

 $\theta$  and  $\phi \rightarrow$  incident and azimuthal angles of electron beam  $d_x$  and  $d_y \rightarrow$  lattice parameters parallel and perpendicular to incident beam

Figure 4.1 Schematic view of the RHEED geometry.  $\Theta_I(\Theta_F)$  and  $\phi_I(\phi_F)$  are the incident and azimuthal angles of the incident (diffracted) beam.  $R_S$  is the distance between substrate and phosphor screen and S the distance between the diffraction spots or streaks.

**RHEED:** Reflection High Energy Electron Diffraction

**Utility of RHEED** 

Determination of vicinal angle

Diffracted intensity → diffraction due to in-plane lattice constant times diff. due to additional periodicity of step-terrace structure.

Out of phase condition → splitting angle

 $\Delta \theta_{f} = (2\pi/kd) (\beta \cos \phi_{p,i}) / (\beta \cos \phi_{p,i} + \langle \theta_{f} \rangle)$ 

 $\beta cos \phi_{p,i} \rightarrow projection of vicinal angle along beam direction$ 

![](_page_42_Figure_8.jpeg)

Figure 4.4 (a) Real space and (b) reciprocal space of a vicinal surface with *a* the in-plane lattice parameter, *d* the step height, and *L* the terrace width. RHEED patterns as recorded from a vicinal SrTiO<sub>3</sub> surface with incident beam perpendicular to the step ledges: (c) in-phase condition and (d) out-of-phase condition.

**RHEED:** Reflection High Energy Electron Diffraction

![](_page_43_Picture_2.jpeg)

+ roughness

#### **RHEED:** Reflection High Energy Electron Diffraction

![](_page_44_Picture_2.jpeg)

#### Layer by layer growth

![](_page_44_Figure_4.jpeg)

![](_page_44_Figure_5.jpeg)

**RHEED:** Reflection High Energy Electron Diffraction

**High pressure RHEED** 

 $I/I_0 = \exp(-I/L_E)$   $L_E \rightarrow$  mean free path,  $I \rightarrow$  path distance

![](_page_45_Figure_4.jpeg)

Figure 4.6 Attenuation  $I/I_0$  of a 10-keV electron beam as a function of the oxygen pressure  $P_{O_2}$  and penetration length 1. The dashed line represents the traveling distance in the high-pressure RHEED setup.

Travelling distance has to be minimized

**RHEED:** Reflection High Energy Electron Diffraction

#### **High pressure RHEED**

Differential pumping system. The travelling distance is reduced to 100mm  $(I/I_0 \sim 0.01)$ 

![](_page_46_Figure_4.jpeg)

![](_page_46_Figure_5.jpeg)

**RHEED:** Reflection High Energy Electron Diffraction

Growth kinetics  $\rightarrow$  PLD is a non-equilibrium technique and kinetics effect could be relevant. High supersaturation  $\rightarrow$  large nucleation rates  $\rightarrow$  kinetic effects control growth mode

Homoepitaxy  $\rightarrow$  2D growth modes are expected (no lattice misfit and thermal coefficient differences). Relevant parameters: surface difussion coeficient (D<sub>s</sub>), sticking coeficient of adatom to a terrace, energy barrier for adatoms to descend to a lower terrace (E<sub>s</sub>).

 $I_D = (D_S \tau)$ , where  $I_D$  is surface diff. lenght and  $\tau$  is reevap. time

 $D_s = va^2 exp(-E_A/k_BT)$ ,  $E_A$  activation energy

The difussion is controlled by the temperature!!

Two processes should be considered: intralayer mass transport and diffusion to a lower terrace

![](_page_48_Figure_0.jpeg)

![](_page_48_Figure_1.jpeg)

000

Slow intralayer mass transport  $\rightarrow$  nucleation at terraces. If interlayer transport is high  $\rightarrow$  layer by layer growth. Otherwise, 3D islands are formed

#### **RHEED:** Reflection High Energy Electron Diffraction

Homoepitaxial SrTiO<sub>3</sub>

![](_page_49_Figure_3.jpeg)

Figure 8.2 Specular RHEED intensity oscillations during homoepitaxial growth of SrTiO<sub>3</sub> at (a) 3 Pa, 850°C, (b) 750°C, and (c) 650°C. The insets in (b) show the enlarged intensities after a laser pulse at different coverage.

Layer-by-layer growth. RHEED relaxations related to adatoms diffusion  $I \sim I_0(1-\exp(-t/\tau))$ , diffusion times, activation energies can be obtained

**RHEED:** Reflection High Energy Electron Diffraction

Homoepitaxial SrTiO<sub>3</sub>

Transition from 2D to step flow (constant RHEED intensity) → adatom diffusion lenght

![](_page_50_Figure_4.jpeg)

Figure 8.7 Specular RHEED intensity variations during homoepitaxial step flow growth of  $SrTiO_3$  at oxyge deposition pressure of (a) 3 Pa and 800°C and (b) at 20 Pa and 875°C. Relaxation times obtained from a fit with Eq. (8.13) for (c) different temperature and oxygen deposition pressure and (d) the same values in the Arthenia form.

#### **RHEED:** Reflection High Energy Electron Diffraction

Atomically smooth surfaces are desired → layer-by-layer growth. Roughening is usually observed (especially at low temperatures and high deposition pressures), related to limited interlayer mass transport. How to avoid this problem? → fast deposition of the amount of material necessary to complete a unit cell in a short interval (supersaturation is achieved for longer times)

![](_page_51_Figure_3.jpeg)

Figure 8.8 Specular RHEED intensity recorded during homoepitaxial growth of SrTiO<sub>3</sub> at 10 Pa and 80 using (a) "standard" PLD and (b) interval PLD. Intensity maxima using (c) "standard" PLD,  $\bigtriangledown$ , and interval PLD,  $\blacktriangle$  (d) Intensity variation during one deposition interval. The surface morphologies of ~30-nm-1 \$\tilde{TIO\_3} films are depicted in the atomic force microscopy (AFM) micrographs (1 × 1 µm<sup>2</sup>): (e) "standard" and (f) interval PLD.

# Remarks

• The importance of thin films and their applications in current research was shown

- Approach to different thin films growth techniques
- Growth mechanisms
- In-situ characterization
- Examples of applications: influence of strain effects on functional properties (magnetic, transport), ReRAM devices, etc.

# <u>Laboratorio de Ablación Láser –</u> <u>CAC - CNEA</u>

# Líneas de Trabajo

- Memorias resistivas (memorias ReRAM)
- Oxidos ferroeléctricos
- Gases electrónicos 2D en sup. e interfaces
- LiMn<sub>2</sub>O<sub>4</sub> para electroquímica

# ¿Qué hacemos?

- Crecimiento`por PLD+RHEED
- XRD, AFM, SEM
- Espectroscopías XPS y Raman
- Micro y nanofabricación en Sala Limpia
- Caracterización eléctrica y magnética
- Modelado

![](_page_53_Figure_13.jpeg)

![](_page_53_Figure_14.jpeg)

**Producción: 20 publicaciones en los últimos 5 años** 

<u>Formación de recursos humanos</u>: I. Alposta (Labo 6&7 UBA, 2012), A. Román (Tesis Lic. UBA 2016), W.R. Acevedo (Doctorado UNSAM, en curso), M. Rengifo (Doctorado UNSAM, en curso), C. Ferreyra (Lic. Física UBA, 2015 y Doctorado UBA, en curso)

<u>Colaboraciones</u>: V. Vildosola & A.M. Llois (CAC), M.J. Sánchez y S. Bengió (CAB), Univ. Groningen (Holanda), Imperial College London (UK), Institute CRISMAT (Francia), nanoGUNE (España)

## Proyectos de Licenciatura y Doctorado. Contacto: diego.rubi@gmail.com